## Time-Dependent Probabilistic Failure of Coated Components

Brice N. Cassenti\* United Technologies Research Center, East Hartford, Connecticut 06108

A probabilistic time-dependent failure model has been developed for the failure of materials. The model is an extension of a previously developed theory for static probabilistic failure that includes fatigue and creep rupture failure. The model has been applied to coatings and ceramic matrix composities. It has been shown to predict accurately the segment size in coatings and the statistical failure locations in brittle tensile specimen. Many of the features of the model can be illustrated using one-dimensional (i.e., through thickness) analyses of layered materials. Such analyses have direct applications to coated components. The model has been incorporated in a computer program that processes output from a finite element simulation and, hence, can be applied to structural analyses, with arbitrary loading, in any number of dimensions.

#### Introduction

OT section gas turbine components are becoming more dependent on ceramic materials to satisfy ever increasing thermal requirements. Thermal barrier coatings and ceramic combustor liners are examples of the uses for high temperature ceramic materials. Although such materials can withstand extremely high temperatures, they are brittle and, hence, possess considerable scatter in their strength. At United Technologies Research Center (UTRC) a probabilistic model has been developed for the failure of materials. The model includes static, creep, and fatigue failure. The static model was developed previously at UTRC1 and has been applied to the analysis of composite material components, solid rocket motor cases, 2 and the prediction of failure locations in brittle specimen tests.3

The model is based on the weakest link failure theory of Weibull.<sup>4,5</sup> Such models make direct use of statistical failure data and, hence, convert deterministic failure analyses into more realistic probabilistic failure analyses. Weibull's original theory predicts only the scatter in static failure loads, and does not predict either the scatter in failure location or failure in time-dependent situations such as fatigue or creep rupture. Deterministic models for time-dependent failure have been the subject of a significant amount of research including fracture, crack propagation, and fatigue. More recently, damage based models<sup>6</sup> have also been introduced. Much of this previous research can be cast directly into a probabilistic form. In fact, fracture mechanics has been viewed from the context of statistical formulations<sup>7-9</sup> and can be used as a basis for more realistic models. Life prediction models for coated components is currently the subject of investigation<sup>10</sup> and will be the major focus of this paper.

#### **Review of Theory**

The theory described in Ref. 1 begins by assuming that the probability of failure  $\delta f$  occurring in an infinitesimal volume  $\delta V$  is given by

$$\delta f = \psi \delta V \tag{1}$$

The parameter  $\psi$  is taken to be a function of the stress  $\sigma_i$ . This dependence can be implicit, e.g.,

$$G(\sigma_i, \psi) = 0,$$
  $i = 1, 2, ..., 6$  (2)

Equation (2) is generally empirical and can take on many forms. One that has been used by the author is

$$G(\sigma_{i}, \psi) = 1 - \sum_{i=1}^{6} \left\langle \frac{\sigma_{i}}{(\psi V_{o})^{1/N_{TI}} \sigma_{T_{i}}} \right\rangle^{M_{T_{i}}} + \left\langle \frac{-\sigma_{i}}{\sigma_{c_{i}} (\psi V_{o})^{1/N_{ci}}} \right\rangle^{M_{ci}} = 0$$
(3)

where

$$\langle x \rangle = \begin{cases} o & x \le 0 \\ x & x > 0 \end{cases}$$

is the unit ramp function,  $\sigma_{1,2,3}$  are normal stress components,  $\sigma_{4,5,6}$  are shear stress components,  $\sigma_{T_i}$ ,  $N_{T_i}$ , and  $M_{T_i}$  are experimentally determined material constants associated with tension failure,  $\sigma_{c_i}$ ,  $N_{c_i}$ , and  $M_{c_i}$  are the equivalent constants for compression failure, and  $V_o$  is a reference volume that can be chosen in a convenient manner for a given material but then should remain constant for the material for all other conditions and geometries. Material test data only determine the products

$$\sigma_{T_i} V_o^{1/N_{T_i}}$$
 or  $\sigma_{c_i} V_o^{1/N_{c_i}}$ 

A more complete discussion of the formulation and interpretation can be found in Ref. 1. Equation (3) can be simplified by considering loading in uniaxial tension only. Then

$$G = 1 - \left[ \frac{\sigma_1}{\sigma_{T_1} (\psi V_o)^{1/N_{T_1}}} \right]^{M_{T_1}} = 0$$
 (4)

Solving for the failure parameter  $\psi$ 

$$\psi = \frac{1}{V_o} \left( \frac{\sigma_1}{\sigma_{T_1}} \right)^{N_{T_1}} \tag{5}$$

Equations (1) and, (2) are not enough to specify failure in an actual component, and, hence, additional assumptions are required. The most conservative assumption would be to assume that if any infinitesimal volume  $\delta V$  fails then the whole structure of volume V fails. This assumption forms the basis of weakest link theory. Applying this assumption to Eq. (1) can be shown to yield the following expression for the probability of survival S of a structure, or component, of volume  $V^1$ 

$$S = \exp\{-\int_{\mathcal{X}} \psi(V) \, dV\}$$
 (6)

Received Aug. 22, 1989; revision received Jan. 2, 1990. Copyright © 1989 by United Technologies Corporation. Published by the American Institute of Aeronautics and Astronautics, Inc., with permission. \*Principal Engineer. Senior Member AIAA.

From a consideration of differential forms of the failure law, Eq. (1) can be shown to result in a differential expression for the probability to observe a failure  $\delta f^{\circ}$  in an infinitesimal volume  $\delta V$ , as 1

$$\frac{\delta f^{\circ}}{\delta V} = \sum_{k=1}^{N} \int_{0}^{P_{k}} S(P') \frac{\partial \psi}{\partial P'_{k}} dP'_{k} = \int_{0}^{t} S(t') \frac{\partial \psi}{\partial t'} dt'$$
 (7)

where  $P_k'$  is one of N loads (e.g., pressure, point loads, etc.) acting on the structure,  $P_k$  is the maximum value of the load  $P_k'$ , and S(P') is the probability of survival for the structure with all the loads acting simultaneously. Note that  $\delta f^\circ$  is the infinitesimal probability to observe a failure at a particular time in the loading history, and  $\delta f$  is the infinitesimal probability for failure to occur at the same point in the loading history. If the structure has already failed,  $\delta f$  cannot be observed. For uniaxial tension, with the distribution of Eq. (5), Eq. (6) yields

$$S = \exp\left\{-\int \left(\frac{\sigma_1}{\sigma_{T_*}}\right)^{N_{T_1}} \frac{\mathrm{d}V}{V_0}\right\} \tag{8}$$

For the case of normal tension loading in the one direction, using Eq. (5), a simple two-parameter Weibull distribution is commonly used. A three-parameter Weibull distribution can also be used but will not be applied in this paper. The integration can be carried out over the volume and the surface to include surface flaw sensitivity.

Equation (7) can be integrated over the volume to find the total probability for failure F,

$$F = \int_{V} \frac{\delta f^{\circ}}{\delta V} \, \mathrm{d}V \tag{9}$$

It can be shown that the sum of S in Eq. (6) and F in Eq. (9) is 1 (see the Appendix), that is,

$$F + S = 1 \tag{10}$$

The remainder of the paper will extend the previous static failure analysis to time-dependent failure and then will apply the results to coated components. Only volumetric flaws will be considered.

## Static Failure Examples

Consider a uniaxial stress  $\sigma$  in a tensile specimen. For a two-parameter Weibull distribution, the failure parameter is

$$\psi(\sigma) = A \, \sigma^m \tag{11}$$

where A and m are material parameters. The parameter A can be chosen to be

$$A = 1/(V_o \sigma_o^m) \tag{12}$$

where  $V_o$  is a characteristic volume and  $\sigma_o$  a characteristic strength.

Recall that the material test data determine the parameter A only. The parameter m in Eq. (11) represents the scatter in failure loads. Higher values of m represent a more deterministic system, whereas lower values of m represent more scatter in the failure loads. If m is infinite, the failure loads would be deterministic. Either one of the parameters  $V_o$  or  $\sigma_o$  in Eq. (12) can be chosen arbitrarily, and the remaining parameter can then be determined by Eq. (12). For a uniform tensile specimen, Eq. (7) can be used to find the failure location distribution and results in

$$\frac{\delta f^{\circ}}{\delta V} = \frac{1 - S}{V} \tag{13}$$

where V is the volume of the specimen and the probability for

survival is

$$S = \exp \left[ -\left(\frac{\sigma}{\sigma_0}\right)^m \frac{V}{V_0} \right] \tag{14}$$

Integrating Eq. (13) over the volume gives

$$F = \int_{0}^{\infty} \frac{\delta f^{\circ}}{\delta V} \, dV = 1 - \exp\left[-\left(\frac{\sigma}{\sigma_{o}}\right)^{m} \frac{V}{V_{o}}\right]$$
 (15)

Adding Eq. (14) and (15)

$$S + F = 1 \tag{16}$$

which must hold.

In a similar manner, the uniaxial case of pure bending can be examined. In pure bending the stress is given by

$$\sigma = \sigma_a(2z/h) \tag{17}$$

where  $\sigma_a$  is the stress at the outside surface, h the distances between surfaces, and z the distance from the centerline.

Assuming only tensile stresses cause failure

$$\psi = \begin{cases} \frac{1}{V_o} \left( \frac{2z \sigma_a}{h \sigma_o} \right)^m & z > 0 \\ 0 & z < 0 \end{cases}$$
 (18)

from Eqs. (11), (12), and (17).

Substituting Eq. (18) into Eq. (12)

$$S = \exp\left\{-\frac{V}{2(m+1)V_o} \left(\frac{\sigma_o}{\sigma_o}\right)^m\right\} \tag{19}$$

Comparing Eqs. (14) and (19), a tensile specimen is more likely to fail at the stress  $\sigma_a$  than a pure bend specimen subject to the same maximum stress  $\sigma_a$ . This results from the fact that much less volume is subject to high stresses in the pure bend specimen than in the tensile specimen.

Another common loading situation is that of three point bending. For this case, the stress is given by (see Fig. 1)

$$\sigma = 4\sigma_a \left(\frac{z}{h}\right) H(x) = 4\sigma_a \left(\frac{z}{h}\right) \begin{cases} \frac{x}{\ell} & \emptyset \le x \le \frac{\ell}{2} \\ \frac{\ell - x}{\ell} & \frac{\ell}{2} \le x \le \ell \end{cases}$$
 (20)

the survival probability is

$$S = \exp\left\{-\frac{V}{2(m+1)^2 V_o} \left(\frac{\sigma_a}{\sigma_o}\right)^m\right\} \tag{21}$$

and the failure location density is

$$\frac{\delta f^{\circ}}{(\delta x/\ell)} = 2^{m}(m+1)H^{m}(x)[1-S]$$
 (22)

If all of the specimens are tested to failure (i.e., S = 0)

$$\frac{\delta f^{\circ}}{(\delta x/\ell)} = 2^{m}(m+1)H^{m}(x) \tag{23}$$

which is illustrated in Fig. 1.

As an example of a coating application, consider the segment size of coatings after fabrication. Often after fabrication, coatings will form a mud-flat cracking pattern. For the purposes of the analysis, assume the following: 1) the coating breaks into hexagonal segments of side length a and a0) the stress state is driven by the substrate and due to the thermal mismatch between the substrate and the coating, then

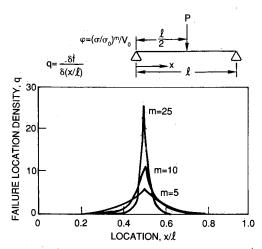


Fig. 1 Three-point bending failure location (taken from Ref. 1).

$$\sigma_x = \sigma_y = \frac{\alpha E \Delta T}{1 - \nu} \tag{24}$$

where  $\sigma_x$  and  $\sigma_y$  are the in-plane stresses,  $\alpha$  the difference in the coefficients of thermal expansion, E and  $\nu$  the elastic constants,  $\Delta T$  the drop in temperature from the glass transition temperature to room temperature, and  $\sigma_x$  and  $\sigma_y$  the in-plane normal stresses, and all other stresses vanish.

Then using the multiaxial failure model, Eq. (4), for an isotropic material and applying Eq. (14) gives for the survival probability

$$1 - F = S = \exp\left\{-\frac{3\sqrt{3}}{2} \left(\frac{a^2 h}{V_o}\right) \left[\frac{\sqrt{2} \alpha E(\Delta T)}{(1 - \nu)\sigma_o}\right]^m\right\}$$
 (25)

where h is the coating thickness. The probability of the segment size to be between a and a + da is dF and

$$\frac{\mathrm{d}F}{\mathrm{d}(ka)} = (ka)e^{-\frac{1}{2}(ka)^2} \tag{26}$$

where

$$k^{2} = 3\sqrt{3} \left(\frac{h}{V_{0}}\right) \left[\frac{\sqrt{2} \alpha E(\Delta T)}{(1 - \nu)\sigma_{0}}\right]$$
 (27)

The probability distributions are illustrated in Fig. 2.

#### Time-Dependent Failure

If failure occurs randomly over the volume (i.e., space), as well as time, then the infinitesimal probability for the failure in the infinitesimal volume  $\delta V$ , and time  $\delta t$ , should be represented by

$$\delta f = \psi \delta V \delta t \tag{28}$$

where

$$(\dot{t}) = \frac{\partial (t)}{\partial t}$$

The parameter  $\psi$  is again a function of the state of the material in  $\delta V$  at time t and will be shown to be a time derivative. The function, though, will not be the time derivative of  $\psi$  in Eq. (1). Then assuming any point failing at any time causes the entire structure to fail (i.e., weakest link theory is assumed to hold), the probability for survival S is given by the product of all of the probabilities for survival

$$S = \prod_{k,l} (1 - \delta f_{k,l}) = \prod_{k,l} \delta_{S_{k,l}}$$
 (29)

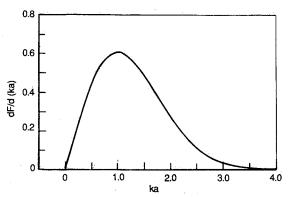


Fig. 2 Segment size density distribution.

where  $\delta_{\mathrm{S}_{k,l}}$  is the probability for survival for element  $\delta V_k$  over time interval  $\delta_{l^k}$ 

Substituting Eq. (28) into Eq. (29) gives

$$S = \prod_{k,l} (1 - \dot{\psi} \delta V_k \delta t_l)$$
 (30)

Taking the natural logarithm of Eq. (30) turns the product into a sum, and

$$lnS = \sum_{k,l} ln \left( 1 - \dot{\psi} \delta V_k \delta t_l \right)$$
 (31)

Since

$$ln(1+x) \approx x$$

for small x, Eq. (31) is

$$\ln S = -\sum_{k,l} \dot{\psi} \delta V_k \delta t_l \tag{32}$$

As the volumes and time intervals each approach zero, the sum becomes an integral, or

$$lnS = -\int_{V} \int_{0}^{t} \dot{\psi} dt dV$$
 (33)

which is exactly analogous to Eq. (12). Then,

$$S = \exp\left\{-\int_{V} \int_{0}^{t} \dot{\psi} dt dV\right\}$$
 (34)

The determination of the failure distribution densities can be found by noting that, in order to observe a failure in interval  $\delta t$  and volume  $\delta V$ , the structure must have survived to time t. Then,

$$\delta f^{\circ} = S \delta f = S \psi \delta V \delta t \tag{35}$$

where  $\delta f^{\circ}$  is the infinitesimal probability to observe a failure in interval  $\delta t$  and volume  $\delta_{V}$ . Integrating over time

$$\frac{\delta f^{\circ}}{\delta V} = \int_{0}^{t} S \dot{\psi} \, dt \tag{36}$$

Several approaches can be used to find functional forms for  $\psi$  in Eq. (28). They include the following: 1) simply taking the parameters in the distribution to be time (or cycle) dependent, <sup>11</sup> 2) using existing damage models, <sup>6</sup> 3) using fatigue or creep rupture data directly. In the first choice, the characteristic strength  $\sigma_o$  in Eq. (12) in a Weibull distribution could be made time dependent. If the exponent m in Eq. (11) is also time dependent, then probabilities for survival would not always shrink. Hence, only the characteristic strength can change with time. For this case, the equations for static failure can be used directly. Then,

$$S(t) = \exp\left\{-\int_{u}^{u} \left[\frac{\sigma(t)}{\sigma_{o}(t)}\right]^{m} \frac{\mathrm{d}V}{V_{o}}\right\}$$
(37)

and

$$\frac{\delta f^{\circ}}{\delta V} = \int_{0}^{t} S(t') \left\{ m \left[ \frac{\sigma(t')}{\sigma_{o}(t')} \right]^{m-1} \dot{\sigma}(t') dt' \right\}$$
(38)

This has been applied to experimental components and found to be accurate. 11

The second option is to apply damage theories. As an example, the damage theory of Chabache will be used, as presented in Ref. 6. In this theory, a damage parameter D is assumed to vary between zero where there is no accumulated damage and 1 when failure at a point has occurred. Both creep damage  $D_c$  and fatigue damage  $D_f$  are considered, and

$$D = D_c + D_f \tag{39}$$

The creep damage is governed by

$$\dot{D}_c = A \left[ \frac{\sqrt{3J_2}}{1 - D_c} \right]^r (1 - D_c)^{r - k} \tag{40}$$

$$k = k_o \exp\left(\mu\sqrt{3j_2}\right) \tag{41}$$

where  $D_c = \mathrm{d}D_c/\mathrm{d}t$ ,  $A,r,k_o$ , and  $\mu$  are material parameters, and  $J_2$  is the second invariant of the deviatoric stress tensor. For a uniaxial stress  $\sigma$ ,

$$J_2 = \frac{1}{3} \sigma^2 \tag{42}$$

The fatigue damage grows per cycle N according to

$$\frac{dD_f}{dN} = [1 - (1 - D_f)^{\beta + 1}]^a \left[ \frac{\Delta \sigma_{II}}{M_o (1 - D_{\sigma_I})(1 - D_f)} \right]^{\beta}$$
(43)

where a,  $\beta$ ,  $M_o$ , and b are material parameters, which may depend on  $\Delta\sigma_H$  abd  $\sigma_B$ 

$$\sigma_{I} = \left[ \left( \frac{\sigma_{1}^{\max} + \sigma_{1}^{\min}}{2} \right) + \left( \frac{\sigma_{2}^{\max} + \sigma_{2}^{\min}}{2} \right) + \left( \frac{\sigma_{3}^{\max} + \sigma_{3}^{\min}}{2} \right) \right] (44)$$

$$\Delta \sigma_{II} = \left[ \frac{\left\{ \left( \sigma_{1}^{\max} - \sigma_{1}^{\min} \right) - \left( \sigma_{2}^{\max} - \sigma_{2}^{\min} \right) \right\}^{2}}{2} + \frac{\left\{ \left( \sigma_{2}^{\max} - \sigma_{2}^{\min} \right) - \left( \sigma_{3}^{\max} - \sigma_{3}^{\min} \right) \right\}^{2}}{2} + \frac{\left\{ \left( \sigma_{3}^{\max} - \sigma_{3}^{\min} \right) - \left( \sigma_{1}^{\max} - \sigma_{1}^{\min} \right)^{2} \right\}^{2}}{2} + \frac{\left\{ \left( \sigma_{3}^{\max} - \sigma_{3}^{\min} \right) - \left( \sigma_{1}^{\max} - \sigma_{1}^{\min} \right)^{2} \right\}^{2}}{2} \right]^{1/2}$$

$$(45)$$

where  $\sigma_i^{\text{max}}$  and  $\sigma_i^{\text{min}}$  are the maximum and minimum *i*th principal stress values during each fatigue cycle, respectively.

A straightforward conversion can be accomplished by taking

$$D_c = 1 - e^{-a_c \psi_c} \tag{46}$$

$$D_f = 1 - e^{-af\psi f} \tag{47}$$

where  $\psi = \psi_f + \psi_c + \psi_s$ , and  $\psi_s$  represents static failure. Equations (46) and (47) can be substituted directly into Eq. (40) and (43) where the time derivatives are simply

 $\dot{D}_c = a_c \dot{\psi}_c \ e^{-a_c \psi_c} \tag{48}$ 

$$\frac{\mathrm{d}D_f}{\mathrm{d}N} = a_f \psi_f \, e^{-a_f \psi_f} \Bigg| \left( \frac{\mathrm{d}N}{\mathrm{d}t} \right) \tag{49}$$

Since  $N = 2\pi\omega t$ 

$$\frac{\mathrm{d}N}{\mathrm{d}t} = 2\pi\omega$$

which is just the frequency in cycles per unit time. The creep and fatigue failure parameters can be taken as

$$\dot{\psi}_c = V_o f_c(\psi) \tag{50}$$

$$\dot{\psi}_f = V_o f_f(\psi) \dot{W} \tag{51}$$

where

$$f_c(\psi) = \frac{a_c(V_o\psi)^r}{1 + a_1 \exp\{b_1 \psi^{1/s}\}}$$
 (52)

$$f_f(\psi) = a_f (1 - e^{(1 + b_2)b\psi})^{\rho_1} \left\{ \frac{\psi^{g/s}}{e^{b\psi} (1 - b_o \psi)^{1/s}} \right\}$$
 (53)

$$p_1 = 1 - \frac{P_o}{C_1(1 - b_2)} \psi^{C_2/s}$$
 (54)

$$\dot{W} = \begin{cases} \sigma_{ij} \dot{\epsilon}_{ij} & \sigma_{ij} \dot{\epsilon}_{ij} > 0 \\ 0 & \sigma_{ij} \dot{\epsilon}_{ij} < 0 \end{cases}$$

 $\sigma_{ij}$  and  $\epsilon_{ij}$  are the stress and strain tensors. For a uniaxial stress state,

$$\dot{W} = \begin{cases} \sigma \epsilon & \sigma \epsilon > 0 \\ 0 & \sigma \epsilon \le 0 \end{cases}$$

The total failure parameter is now given by summing the fatigue, creep, and static parts. Instead of damage, entropy gain<sup>12</sup> could be used in a similar manner. Using damage, or entropy gain, has the disadvantage of requiring many parameters. The parameters in Eqs. (51–54) must be found by matching experimental data.

The third approach offers the best compromise in terms of simplicity and completeness by using existing fatigue or creep rupture directly. In this approach, it is more difficult to derive the equations for the failure parameter  $\psi$ .

In the modeling of fatigue failure, it is commonly assumed that the number of cycles to failure N and the maximum applied cyclic stress  $\sigma_m$  are related by 12

$$N\sigma_m^a = \text{const}$$
 (55)

If the constant in Eq. (55) is assumed to be a function of the survival probability  $S_N$ , then after N cycles

$$CN^n \sigma_m^m = k(S_N) \tag{56}$$

where m = na and C is a constant. The constant can be taken so that

$$\left(\frac{V}{V_o}\right) \left(\frac{N}{N_o}\right)^n \left(\frac{\sigma_m}{\sigma_f}\right)^m = k(S_N) \tag{57}$$

where  $V_o N_o^n \sigma_f^m$  is a material parameter.

Solving Eq. (57) for  $S_N$ , and assuming a uniform stress state

$$S_N = e^{-\psi_N V} = k^{-1} \left[ \left( \frac{V}{V_o} \right) \left( \frac{N}{N_o} \right)^n \left( \frac{\sigma_m}{\sigma_f} \right)^m \right]$$
 (58)

Taking

$$k^{-1}(x) = e^{-x}$$

Eq. (58) becomes

$$\psi_N = \frac{1}{V_o} \left( \frac{N}{N_o} \right)^n \left( \frac{\sigma_m}{\sigma_f} \right)^m \tag{59}$$

In order to arrive at a differential form for  $\psi$ , we must con-

sider the change in the failure parameter over one cycle

$$\psi_{N+1} - \psi_N = \Delta \psi_N = \frac{1}{V_o} \left[ \left( \frac{N+1}{N_o} \right)^n - \left( \frac{N}{N_o} \right)^n \right] \left( \frac{\sigma_m}{\sigma_f} \right)^m \tag{60}$$

For high cycle fatigue, we only need to consider the case where

then

$$\Delta \psi_N = \frac{n}{V_0 N_0} \left( \frac{N}{N_0} \right)^{n-1} \left( \frac{\sigma_m}{\sigma_f} \right)^m \tag{61}$$

This is equivalent to an integration over the cycle while the stress is positive,

$$\Delta \psi_N = \int_{\sigma=0}^{\sigma=\sigma_m} d\psi_f = \frac{nm}{V_o N_o \sigma_o} \left(\frac{N}{N_o}\right)^{n-1} \int_{\sigma=0}^{\sigma=\sigma_m} \left(\frac{\sigma}{\sigma_f}\right)^{m-1} d\sigma \quad (62)$$

Equating the differentials in Eq. (62) gives

$$\dot{\psi}_f = \frac{mn}{V_o N_o} \left(\frac{N}{N_o}\right)^{n-1} \left(\frac{\sigma}{\sigma_f}\right)^{m-1} \frac{\dot{\sigma}}{\sigma_f} \tag{63}$$

But from Eq. (59)

$$\frac{N}{N_o} = \left[ \frac{V_o \psi_N}{(\sigma_m / \sigma_f)^m} \right]^{1/n} = \left[ \frac{V_o \psi_f}{(\sigma_m / \sigma_f)^m} \right]^{1/n} \tag{64}$$

In Eq. (64), the maximum stress can be replaced by the current stress if a multiplicative constant K is introduced, which will be determined later. Then if  $\Delta\psi_N$  is much less than  $\psi_N$  we can set

$$\frac{N}{N_o} = K \left[ \frac{V_o \psi_f}{(\sigma/\sigma_f)^m} \right]^{1/n} \tag{65}$$

substituting Eq. (65) into Eq. (63) gives

$$\dot{\psi}_f = \frac{mnK}{V_o N_o} \left( V_o \psi_f \right)^{1 - \frac{1}{n}} \left\langle \frac{\sigma}{\sigma_f} \right\rangle^{m - 1} \left\langle \frac{\dot{\sigma}}{\sigma_f} \right\rangle \tag{66}$$

Equation (66) will be taken to be the equation governing the growth in the failure parameter, which only occurs when both  $\sigma$  and  $\sigma$  are greater than zero. Recall  $\langle x \rangle$  is the unit ramp function. Equation (66) can be integrated over one cycle with the initial condition

$$\psi_f = \psi_N$$
 at  $\sigma = 0$ 

then

$$\psi_{N+1} = \frac{a}{V_o} \left[ (V_o \psi_N)^{1/n} + \frac{Kn}{N_o} \left( \frac{\sigma}{\sigma_f} \right)^{m/n} \right]^n \tag{67}$$

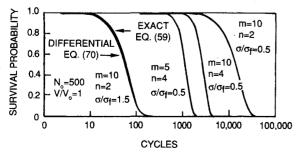


Fig. 3 Fatigue survival probability using exact and differential forms.

Taking

$$K = 1/n \tag{68}$$

will give the correct answer for N = 0.

Then Eqs. (66) and (67) become

$$\psi_f = \frac{m}{V_o N_o} (V_o \psi_f) \left( 1 - \frac{1}{n} \right) \left\langle \frac{\sigma}{\sigma_f} \right\rangle^{\frac{m}{n} - 1} \left\langle \frac{\sigma}{\sigma_f} \right\rangle$$
 (69)

$$\psi_{N+1} = \frac{1}{V_o} \left[ (V_o \psi_N)^{1/n} + \frac{1}{N_o} \left( \frac{\sigma_m}{\sigma_f} \right)^{m/n} \right]^n$$
 (70)

In Fig. 3, Eqs. (59) and (70) are compared. The results are seen to be nearly identical, even for a small number of cycles. Figure 4 compares the uniaxial, Eq. (14), pure bending, Eq. (19), and three point bending load cases, Eq. (21). For the same maximum stresses, the life can vary by an order of magnitude in these three load cases.

A probabilistic failure model for creep rupture can be developed in an analogous manner. Taking

$$\psi_c = \frac{1}{V_o} \left( \frac{\sigma}{\sigma_c} \right)^M \left( \frac{t}{t_o} \right)^N \tag{71}$$

where the stress is now considered a constant. Then

$$\dot{\psi}_c = \frac{N}{V_o t_o} \left(\frac{\sigma}{\sigma_c}\right)^M \left(\frac{t}{t_o}\right)^{N-1} \tag{72}$$

From Eq. (71)

$$\left(\frac{t}{t_o}\right) = (V_o \psi_c)^{1/N} \left(\frac{\sigma}{\sigma_c}\right)^M \tag{73}$$

Substituting Eq. (73) into Eq. (72) and assuming that only positive stresses cause failure

$$\dot{\psi}_c = \frac{N}{V_o t_o} \left\langle \frac{\sigma}{\sigma_c} \right\rangle^{M/N} (V_o \psi_c)^{1 - \frac{1}{N}}$$
 (74)

The failure parameter  $\dot{\psi}$  can now be taken as the sum of Eq. (69) and (74). Static failure can be represented by the maximum value of the static failure parameter over all times, or

$$\psi_s = \max_i \left( \psi_o \right) \tag{75}$$

where  $G(\sigma_i, \psi_o) = 0$  from Eqs. (2) and (3). The total  $\psi$  is

$$\psi = \psi_s + \int_0^t \left[ \dot{\psi}_f(t') + \dot{\psi}_c(t') \right] dt'$$
 (76)

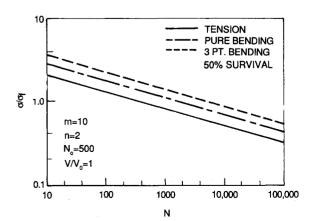


Fig. 4 Fatigue life example.

The evolution Eqs. (74) and (69) can be written as

$$\dot{\psi}_c = \frac{N}{V_o t_o} \left\langle \frac{\sigma}{\sigma_c} \right\rangle^{m/N} (V_o \psi)^{1 - \frac{1}{N}} \tag{77}$$

$$\dot{\psi}_f = \frac{m}{V_o N_o} \left\langle \frac{\sigma}{\sigma_f} \right\rangle^{\frac{m}{n} - 1} (V_o \psi)^{1 - \frac{1}{n}} \left\langle \frac{\dot{\sigma}}{\sigma_f} \right\rangle \tag{78}$$

respectively. The right-hand sides of Eqs. (77) and (78) now contain the total failure parameter instead of the individual creep and fatigue failure parameters. This should be more accurate if both creep and fatigue failure are driven by the same mechanism (e.g., the propagation of cracks along grain boundaries). Equations (75–78) contain the complete uniaxial stress state model that will be applied in the next section. Multiaxial stress states can be described by expressions similar to Eq. (3) and will be briefly examined in the next section.

### Sample Applications

As an example, consider a coating driven by a much stiffer substrate through a strain and thermal cycle. Consider first a one-dimensional case. The coating will be modeled using a glass transition temperature. Above the glass transition temperature the coating cannot support any stresses, and below the glass transition temperature the coating is elastic. The coating strain will be taken to be that of the substrate. Hooke's law, in one dimension, gives for the coating stress

$$\sigma = E \left[ \epsilon - \alpha (T - T_r) \right] \tag{79}$$

where  $\sigma$  is the coating stress; E the Young's Modulus of the coating;  $\alpha$  the coefficient of thermal expansion of the coating;

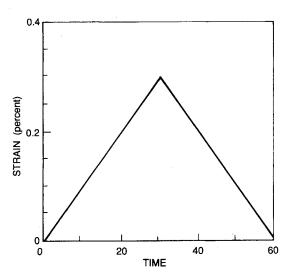


Fig. 5 Strain history for coating.

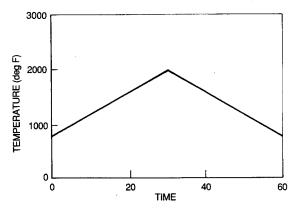


Fig. 6 Temperature history for coating.

T the coating temperature;  $\epsilon$  the coating strain, which is equal to the substrate strain; and  $T_r$  a reference temperature.

The coating reference temperature can be found by noting that, at the glass transition temperature, the stress vanishes. Then

$$\epsilon_g - \alpha \left( T_g - T_R \right) = 0 \tag{80}$$

where  $\epsilon_g$  is the applied substrate strain when the temperature was last equal to the glass transition temperature  $T_g$ . Solving Eq. (80) for  $T_R$  gives

$$T_{R} = T_{g} - \left(\epsilon_{g}/\alpha\right) \tag{81}$$

Substituting Eq. (81) into Eq. (79) completes the stress strain law

$$\sigma = \begin{cases} E\left[\epsilon - \epsilon_g - \alpha(T - T_g)\right] & T < T_g \\ 0 & T \ge T_g \end{cases}$$
(82)

The applied strain and temperature history will be taken as triangular waves, as shown in Figs. 5 and 6. The corresponding stress history is shown in Fig. 7 for the data in Table 1. The stress history can be integrated over one cycle according to Eq. (69) to yield the evolution of the failure parameter  $\psi$  and the corresponding survival probability as

$$\psi_{N+1} = \frac{1}{V_o} \left\{ (V_o \psi_N)^{1/N} + \frac{m}{V_o N_o} \delta \right\}^n$$
 (83)

$$S_{N+1} = e^{-\psi_{N+1}V} \tag{84}$$

where

$$\delta = \int_{\text{cycle}} \left( \frac{\langle \sigma \rangle}{\sigma_f} \right)^{\frac{m}{n} - 1} \left\langle \frac{\dot{\sigma}}{\sigma_f} \right\rangle \, dt \tag{85}$$

The probability of survival as a function of cycle number for the parameters of Table 1, and the case where V and  $V_o$  are the same, is shown in Fig. 8. As a multiaxial example, consider a biaxial stress state in a coating that lies in the x-y

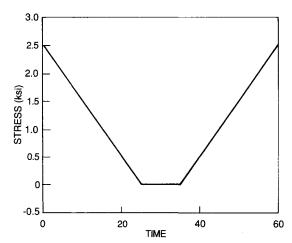


Fig. 7 Coating stress history.

Table 1 Values of parameters for coating example

Parameter	Value
m	10
n	2
$N_o$	500
$\sigma_f$	5000 psi $5 \times 10^{-6} / {}^{\circ}F$
ά	$5 \times 10^{-6}$ °F
$T_g$	1800 F
${E}$	10 <sup>6</sup> psi

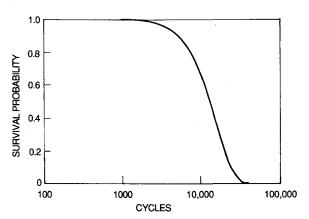


Fig. 8 Survival probability history of coating.

plane. Hooke's law is

$$\epsilon_x = (1/E)(\sigma_x - \nu\sigma_y) + \alpha(T - T_g) + \epsilon_{x_g}$$
 (86)

$$\epsilon_y = (1/E)(\sigma_y - \nu \sigma_x) + \alpha(T - T_g) + \epsilon_{y_g}$$
 (87)

$$\gamma_{xy} = \left(1/G\right)\tau_{xy} + \gamma_{xy_g} \tag{88}$$

where G is the shear modulus,  $\gamma_{xy}$  the coating shear strain,  $\tau_{xy}$  the coating shear stress, and  $\gamma_{xy_g}$  the applied substrate strain at the glass transition temperature.

A multiaxial law is now required for the failure parameter. This can now be taken, in a manner similar to Eq. (3), as

$$\dot{\psi} = \frac{m}{V_o N_o} (V_o \psi) \left( 1 - \frac{1}{n} \right) \left\{ \left( \frac{\langle \sigma_x \rangle}{\sigma_{x_f}} \right)^p \left( \frac{\langle \sigma \rangle}{\sigma_{x_f}} \right)^{p/m} + \left( \frac{\langle \sigma_y \rangle}{\sigma_{x_f}} \right)^p \left( \frac{\langle \sigma_y \rangle}{\sigma_{x_f}} \right)^{p/m} - \left( \frac{|\tau_{xy}|}{\tau_f} \right)^p \left( \frac{|\tau_{xy}|}{\tau_f} \right)^{p/m} \right\}^{m/p}$$
(89)

For the case where the imposed strains are isotropic

$$\epsilon_x = \epsilon_y = \epsilon$$
 (90)

$$\gamma_{xy} = 0 \tag{91}$$

Then

$$\sigma_{\rm r} = \sigma_{\rm v} = \sigma \tag{92}$$

$$\tau_{xy} = 0 \tag{93}$$

Since the properties are isotropic, Eq. (89) becomes

$$\dot{\psi} = 2^{m/p} \left( \frac{M}{V_o N_o} \right) (V_o \psi) \left( 1 - \frac{1}{n} \right) \left( \frac{\langle \sigma \rangle}{\sigma_f} \right)^m \frac{\langle \sigma \rangle}{\sigma_f}$$
(94)

where

$$\sigma_{x_f} = \sigma_{y_f} = \sigma_f \tag{95}$$

Equation (94) is equivalent to Eq. (78) to within a multiplicative constant. Equation (89) can use the principal stresses instead of the actual stress components. Many other forms also need to be examined. For anisotropic materials, more complicated expressions need to be developed.

The model has been added as a postprocessor to a finite element code and has been used to analyze design concepts for an advanced gas turbine engine combustor. The applications have described accurately the results of experimental tests, but in each case, static failure has dominated and, hence, a real test of the model still needs to be done.

#### **Conclusions**

Probabilistic formulations for static failure have been extended to time-dependent failure. Three methods for completing the extension have been described. An example for a multiaxial formulation has been given and an application to uniaxial stress states has demonstrated the results that can be obtained. The model still needs a consistent multiaxial formulation especially for anisotropic materials and needs to be compared to a wider variety of experimental results.

#### Appendix: Sum of Failure and Survival Probabilities

From Eq. (6), the survival probability is

$$S = \exp[-\int \psi \, \mathrm{d}V] \tag{A1}$$

and from Eq. (9), the failure probability is

$$F = \int_{V} \frac{\delta f^{\circ}}{\delta V} \, \mathrm{d}V \tag{A2}$$

where from Eq. (7)

$$\frac{\delta f^{\circ}}{\delta V} = \int_{0}^{t} S(t_1) \psi_{,t_1}(\mathbf{x}, t_1) dt_1$$
 (A3)

where

$$\psi_{,t} = \frac{\partial \psi}{\partial t}$$

Let

$$P(t) = F(t) + S(t) \tag{A4}$$

ther

$$P(t) = \int_{V}^{t} \exp\left[-\int_{V} \psi \, dV\right] \psi_{,t} \, dt_{1} \, dV$$
  
+  $\exp\left[-\int_{V} \psi \, dV\right]$  (A5)

Differentiating Eq. (A-5)

$$\frac{\partial P}{\partial t} = \int_{V} \exp[-\int_{V} \psi \, dV] \psi_{,t} \, dV - \exp[-\int_{V} \psi_{,t} \, dV] \qquad (A6)$$

But

$$\exp\left[-\int_{V}\psi\,\mathrm{d}V\right] = S(t)$$

from Equation (A1) and, hence,

$$\frac{\mathrm{d}P}{\mathrm{d}t} = 0\tag{A7}$$

Then integrating

$$P = C_1 = F + S \tag{A8}$$

where  $C_1$  is an arbitrary constant.

If there are no failures due to residual stresses resulting from fabrication, then

$$S = 1 \qquad \text{and} \qquad F = 0 \qquad \text{at } t = 0 \tag{A9}$$

the arbitrary constant is then 1 and Equation (A-8) becomes

$$F + S = 1 \tag{A10}$$

#### References

<sup>1</sup>Cassenti, B. N., "Probabilistic Static Failure of Composite Material," *AIAA Journal*, Vol. 22, No. 1, 1984, pp. 103-110.

<sup>2</sup>Cassenti, B. N., "Statistical Failure Analyses of Composite Solid Rocket Motor Cases," AIAA Paper 85-1100, June 1987.

<sup>3</sup>Vasko, T. J., and Cassenti, B. N., "The Statistical Prediction of Failure Location in Brittle Test Specimens," Proceedings of the AIAA/ASME/ASCE/AHS 27th Structures, Structural Dynamics and Materials Conference, AIAA, New York, May 1986.

4Weibull, W., "A Statistical Theory of the Strength of Materials," Royal Swedish Academy of Engineering Sciences, No. 151, 1939.

<sup>5</sup>Weibull, W., "A Statistical Distribution Function of Wide Applicability," Journal of Applied Mechanics, Vol. 18, No. 3, 1951, pp.

<sup>6</sup>Wilson, D. A., and Walker, K. P., "Constitutive Modeling of Engine Materials," Air Force Wright Aeronautical Lab., Materials Lab., Wright-Patterson AFB, OH, TR-84-7073, Sept. 1983.

<sup>7</sup>Duan, K., Mai, Y.-W., and Cotterell, B., "A Statistical Fracture Mechanics of Time-Dependent Strength Behavior of Partially Stabilized Zirconia," Journal of Materials Science, Vol. 23, 1988, pp. 3671-3677.

<sup>8</sup>Batdorf, S. B., "Fracture Statistics of Polyaxial Stress States," Fracture Mechanics, Proceedings of the International Symposium on Fracture Mechanics, edited by N. Perrone, H. Liebowitz, D. Mulville, and W. Pilkey. Univ. of Virginia Press, Charlottesville, VA, Sept., 1978, pp. 579-591.

<sup>9</sup>Hu, X.-Z., Mai, Y.-W., and Cotterell, B., "A Statistical Theory of Time-Dependent Fracture for Brittle Materials," Philosophical

Magazine A, Vol. 58, 1988, pp. 299-324.

<sup>10</sup>Curse, T. A., Stewart, S. E., and Ortiz, M., "Thermal Barrier Coating Life Prediction Model Development," Presented at the Gas Turbine and Aeroengine Congress and Exposition, Amsterdam, June

<sup>11</sup>Siskind, K. S., and Able, E. C., private communication, 1987. <sup>12</sup>Whaley, P. W., "A Thermodynamic Approach to Material Fatigue," ASME International Conference on Advances in Life Prediction Methods, Albany, NY, April 1983.

<sup>13</sup>McClintock, F. A., and Argon, A. S., Mechanical Behavior of Materials, Addison-Wesley, Reading, MA, 1966.

Recommended Reading from the AIAA Progress in Astronautics and Aeronautics Series . . . dala-



# **Spacecraft Dielectric Material Properties** and Spacecraft Charging

Arthur R. Frederickson, David B. Cotts, James A. Wall and Frank L. Bouquet, editors

This book treats a confluence of the disciplines of spacecraft charging, polymer chemistry, and radiation effects to help satellite designers choose dielectrics, especially polymers, that avoid charging problems. It proposes promising conductive polymer candidates, and indicates by example and by reference to the literature how the conductivity and radiation hardness of dielectrics in general can be tested. The field of semi-insulating polymers is beginning to blossom and provides most of the current information. The book surveys a great deal of literature on existing and potential polymers proposed for noncharging spacecraft applications. Some of the difficulties of accelerated testing are discussed, and suggestions for their resolution are made. The discussion includes extensive reference to the literature on conductivity measurements.

TO ORDER: Write, Phone or FAX: AIAA c/o TASCO, 9 Jay Gould Ct., P.O. Box 753, Waldorf, MD 20604 Phone (301) 645-5643, Dept. 415 FAX (301) 843-0159

Sales Tax: CA residents, 7%; DC, 6%. For shipping and handling add \$4.75 for 1-4 books (call for rates for higher quanties). Orders under \$50.00 must be prepaid. Foreign orders must be prepaid. Please allow 4 weeks for delivery. Prices are subject to change without notice. Returns will be accepted within 15 days.

1986 96 pp., illus. Hardback ISBN 0-930403-17-7 AIAA Members \$29.95 Nonmembers \$37.95 Order Number V-107